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Improving Tin Whisker Testing through Quantitative Measurements of Plated Film Properties

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Biography:

Aaron Pedigo received his B.S. in Materials Engineering from Purdue in 2006. He continued his education at Purdue by enrolling as a PhD student in Materials Engineering, researching the growth of tin whiskers. In 2009, Aaron began working for NSWC Crane while continuing his graduate studies.

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Improving Tin Whiskers through Quantitative Measurements of Plated Film Properties

Aaron Pedigo^{1,2}, Pylin Sarobol², John Blendell², and Carol Handwerker²
September 29th, 2010



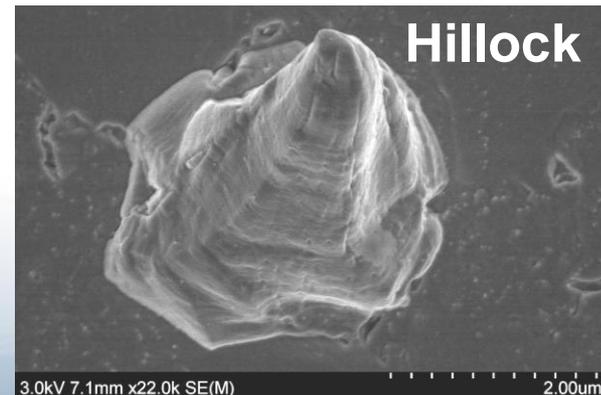
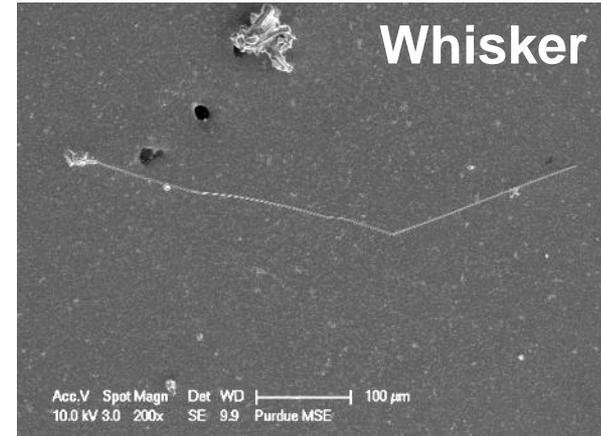
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Introduction: Whiskers and Hillocks

- Tin whiskers grow spontaneously from Sn based finishes
 - Conductive
 - Grow to be millimeters in length
 - **Represents a significant electric reliability risk**
- Mechanisms responsible for whisker growth not well understood
 - Multiple surface growths including hillocks and whiskers
 - **Hinders successful mitigation strategies**

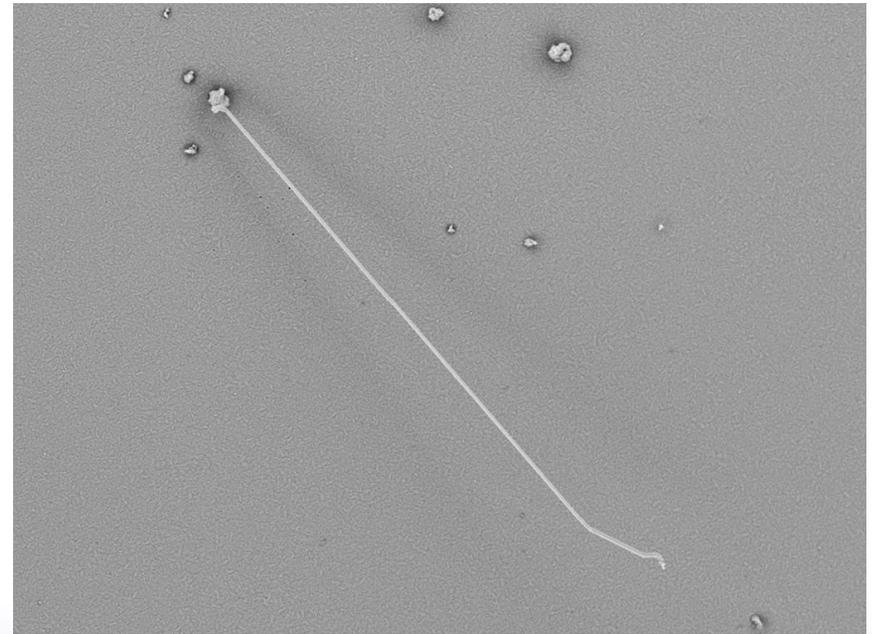


Introduction:

The Pb-Free Manhattan Project

http://www.navyb2pcoe.org/b2p_news_ifmp2.html

- Team of experts from both industry and academia
- Addresses particular concerns of the Aerospace and Defense industries with Pb-Free electronics
- Tin whiskers identified as one of the **“greatest reliability risks associated with lead-free electronics.”**



Cu Tin

2008/11/18

x600 100 um

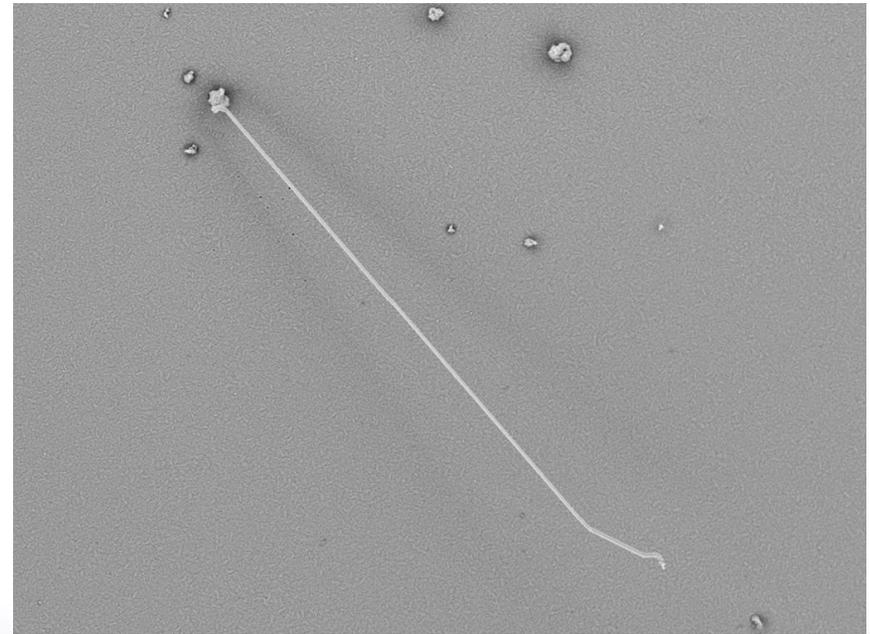
SEM micrograph showing a a 300mm whisker growing from a electroplated Sn finish containing <2% Cu after 1 year.

Introduction:

The Pb-Free Manhattan Project

http://www.navyb2pcoe.org/b2p_news_ifmp2.html

- From Pb-Free Manhattan Project Phase 2 Report:
 - “**Current tin whisker testing methods** cannot predict whether a finish or solder will grow tin whiskers, nor can they identify what additional whisker mitigation strategies are needed for a particular part or assembly.”



Cu Tin 2008/11/18 x600 100 um

SEM micrograph showing a a 300mm whisker growing from a electroplated Sn finish containing <2% Cu after 1 year.

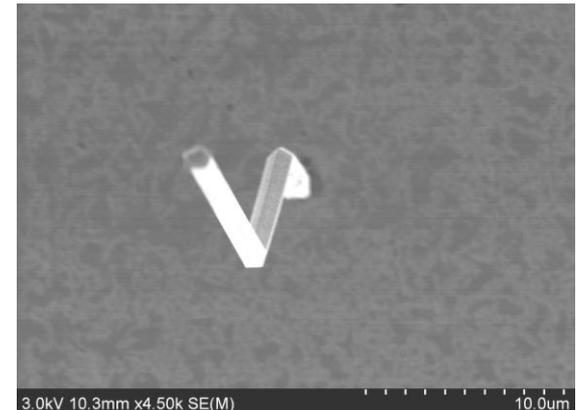
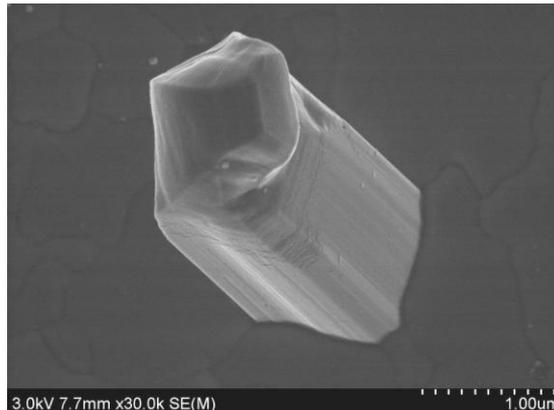
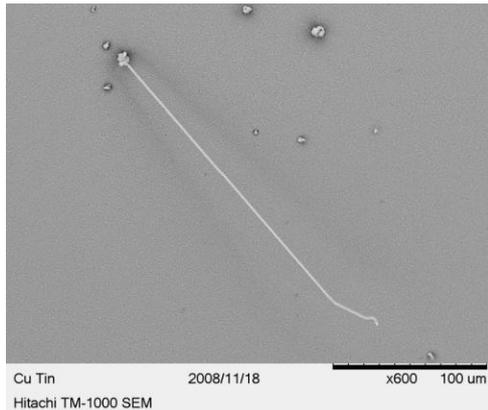
Introduction:

JEDEC standard JESD22A121

- Sn whisker testing standard
 - Variety of of storage conditions
 - Ambient, thermal hold, and thermal cycling
 - Time consuming
 - 3000hrs (125days)
 - Must be repeated if there are changes in the process
 - Electrolyte additive, processing parameters, reflow profile
 - Binary result
 - The finish is or is not prone to whiskering
 - **From JEDEC standard JESD22A121**
 - “[T]here is at present no way to quantitatively predict whisker lengths over long time periods based on the lengths measured in the short-term tests described in this document. At the time of writing, the **fundamental mechanisms** of tin whisker growth are not fully understood and acceleration factors have not been established” - May 2005

Motivation:

Understanding Fundamental Whisker Growth Mechanisms



- Developing successful mitigation strategies
 - Perform experiments to determine film properties related to whisker growth to **develop a comprehensive whisker growth model**

Motivation:

Comprehensive Whisker Growth Model

- A comprehensive whisker growth model is composed of the fundamental mechanisms responsible for whisker growth and should be able to explain empirical observations related to whisker growth
- Empirical Observations
 - Compressive stress driving whisker growth¹
 - Straight whiskers, kinked whiskers, and hillocks all growing on the same sample
 - Surface contamination influencing the propensity to whisker²
 - Bright Sn more susceptible than matte Sn
 - Co-depositing Cu and Sn increases propensity to whisker^{3,4,5,6,7}

Methods:

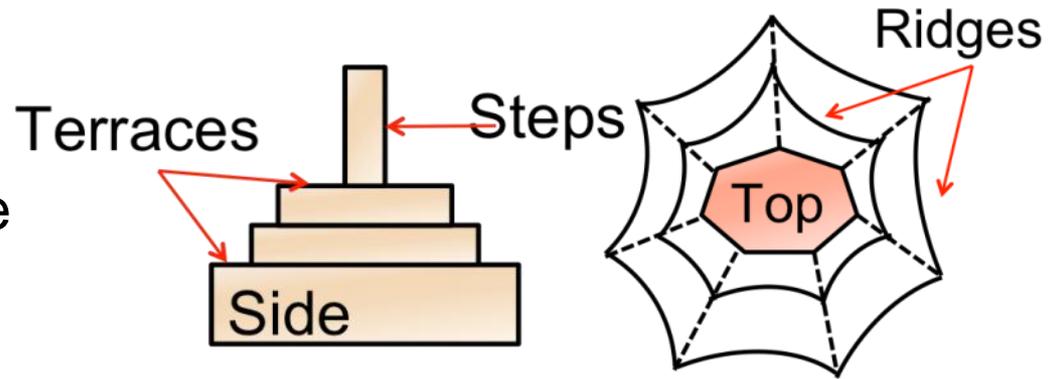
Measurements of Plated Film Properties

- Hillock and Whisker Growth
 - Presence of whiskers and hillocks
 - Morphology of hillocks
- Film Stress Evolution
 - Deposition stress
 - Film stress as a function of time
- Microstructure
 - Crystallographic texture
 - Grain size
 - Grain morphology

Methods:

Hillock and Whisker Growth

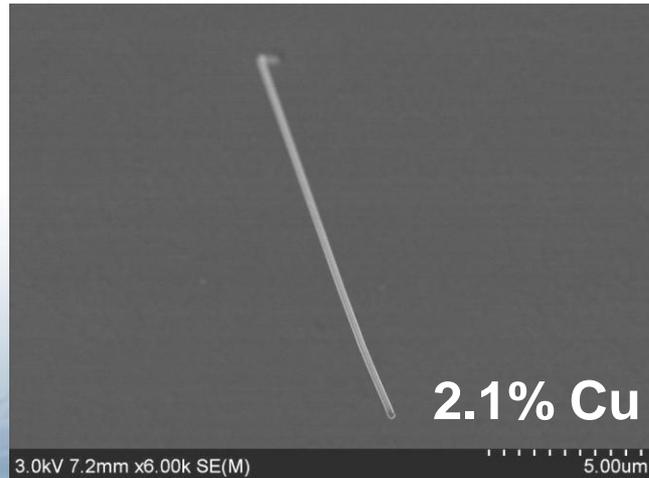
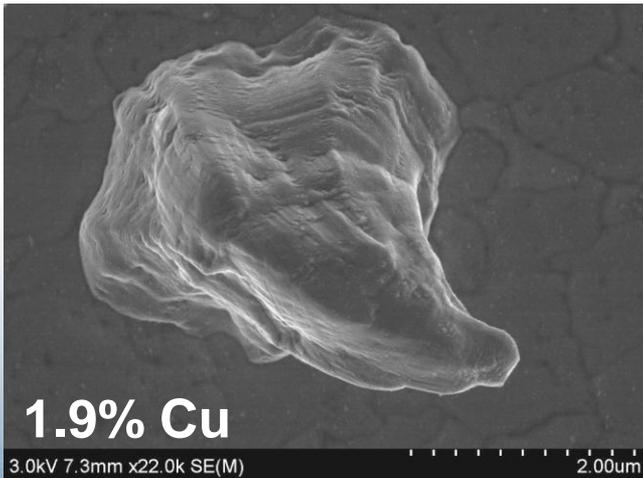
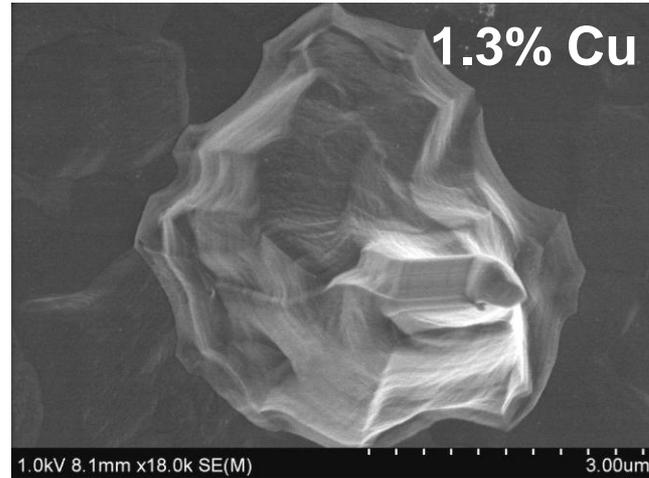
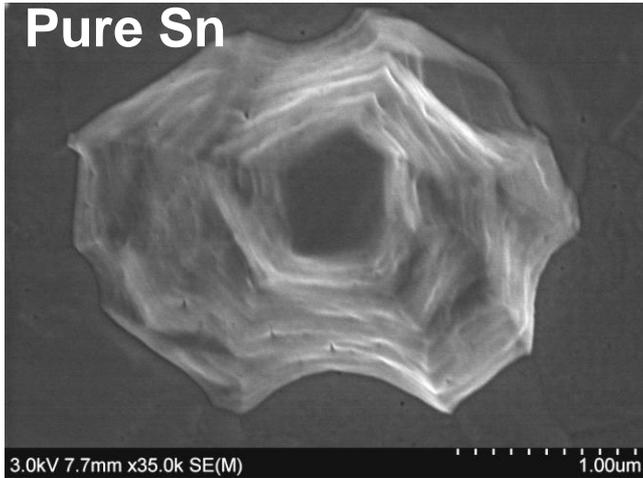
- Electrodeposition of Sn and SnCu films⁵
 - Film ranging in composition from pure Sn to 2.1wt% Cu, Sn
 - Hillock and whisker growth observed a short time after deposition
 - A range in hillock morphologies as a function of Cu content in the film



Schematic of a proposed hillock growth model created from hillock morphology observations.

Results:

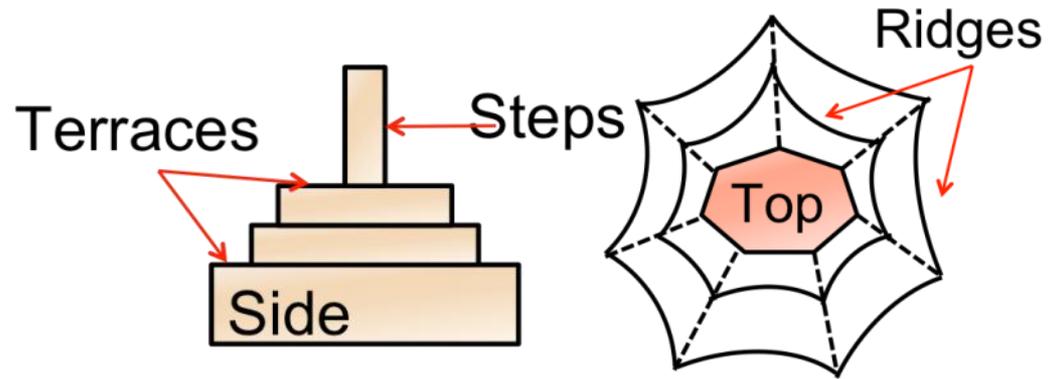
Hillock and Whisker Growth



Results:

Hillock and Whisker Growth

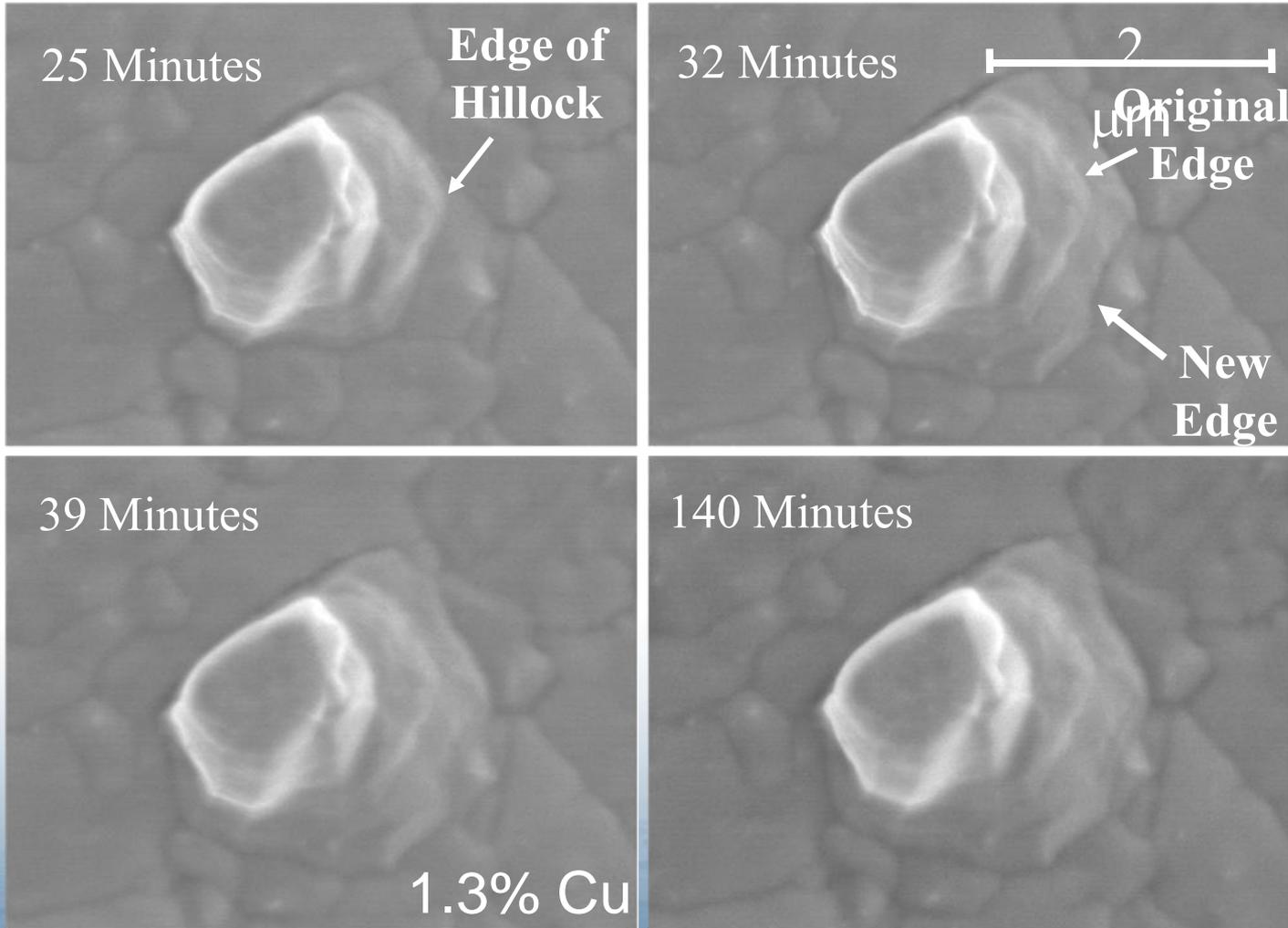
- Increasing Cu concentration resulted in
 - Transition from hillock only to hillock and whisker growth
 - Agrees with previous studies^{1,2}
 - Hillock growth where the rate of uplift increased compared to the rate of lateral growth
 - Hillocks with whisker-like tops



Schematic of a proposed hillock growth model created from hillock morphology observations.

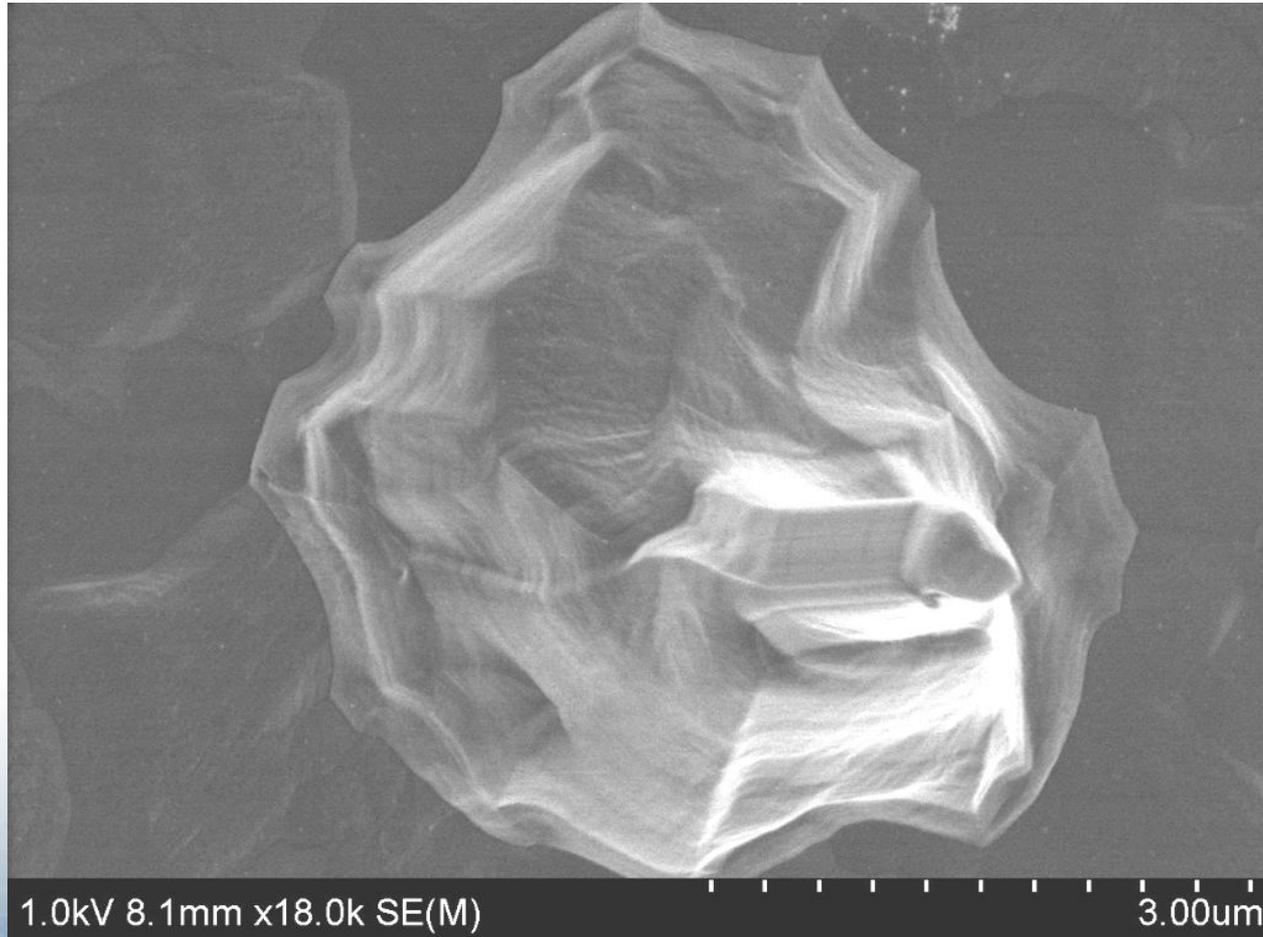
Results:

Hillock and Whisker Growth



Results:

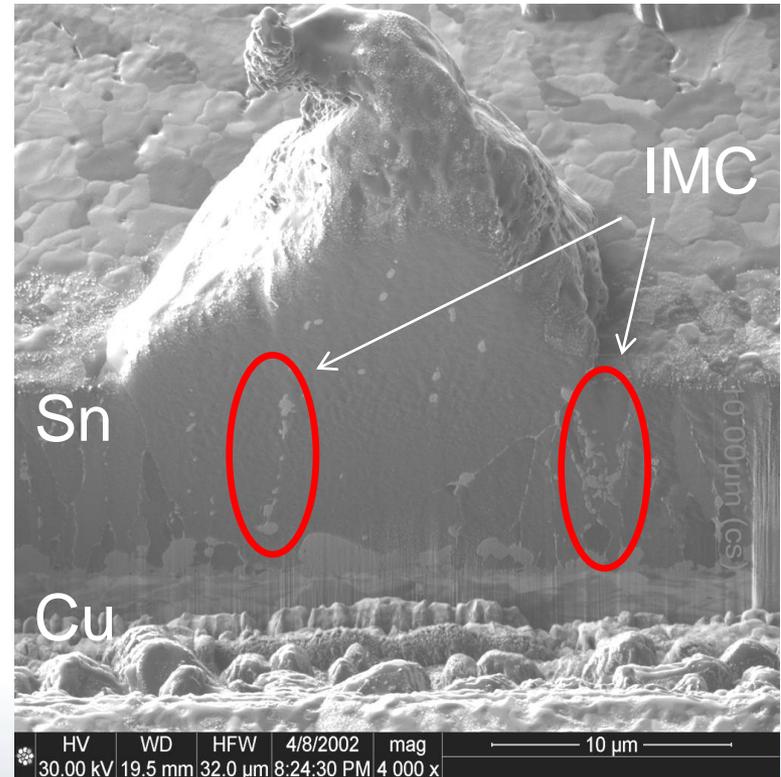
Hillock and Whisker Growth



Results:

Hillock and Whisker Growth

- Mechanism: Grain boundary pinning
 - Co-deposition of Cu and Sn results in Cu_6Sn_5 IMC along grain boundaries (GB). These IMC can decrease GB mobility or completely pin the GB.
 - What is pinning the GB in “pure” Sn deposits?



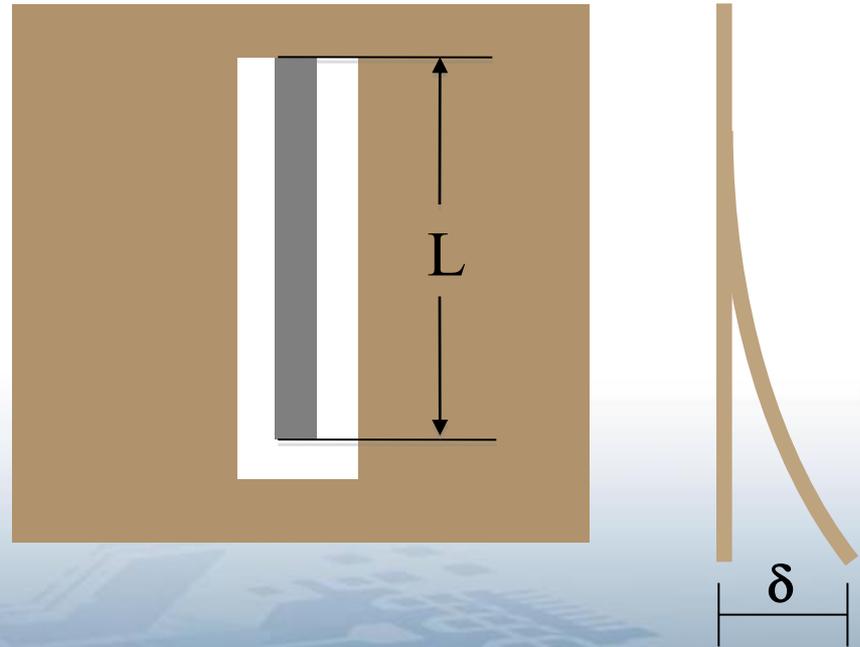
FIB milled cross-section through a hillock showing lines of IMC marking previous locations of GBs.

Methods:

Film Stress Evolution

- Same compositions evaluated for hillocks and whisker observations deposited on flexible cantilever beams
- Deflection of the cantilever beam is related to the film stress (σ_f)⁸
- Measurements made as a function of time

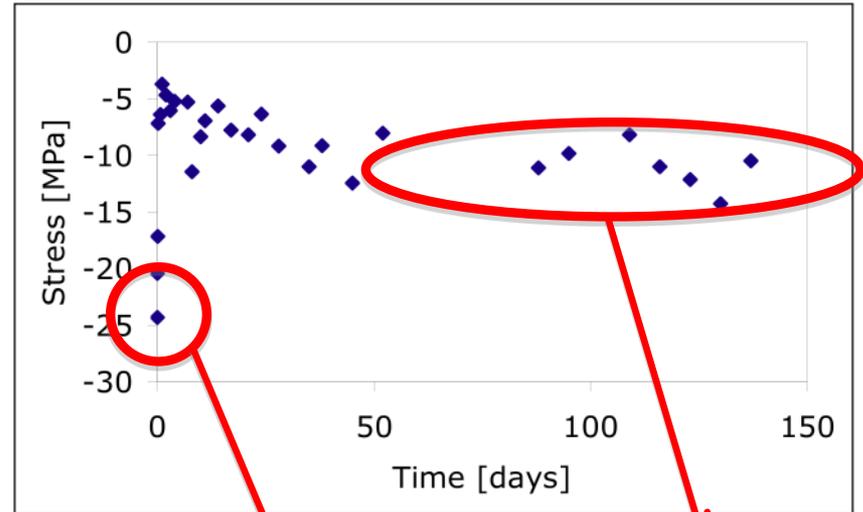
$$\sigma_f = \frac{\text{Modulus}_{\text{Substrate}}}{6(1 - \text{poisson}'s_{\text{substrate}})} \times \frac{\text{thickness}_{\text{substrate}}^2}{\text{thickness}_{\text{film}}} \times \frac{2\delta}{L^2}$$



Results:

Film Stress Evolution

- Same general trends in stress evolution regardless of Cu concentration
- Increasing Cu concentration resulted in
 - Increased compressive plating stress
 - Increased compressive long-term residual stress
- Results agree with previous studies

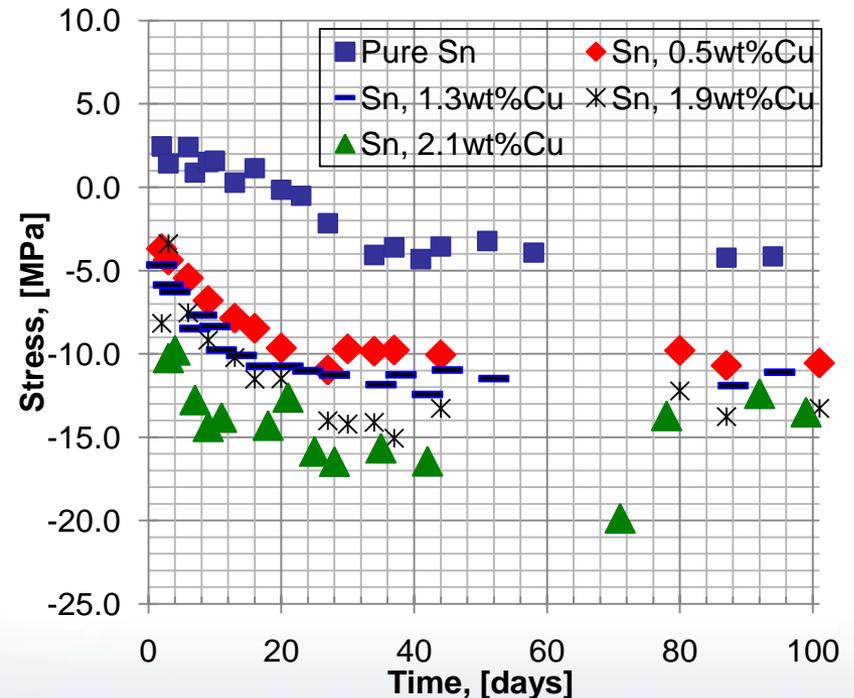


Copper Conc. [%]	Plating Stress [MPa]	30-150 Day Average Stress [MPa]
0.0	-17 (+/- 4)	-7 (+/- 3)
0.5	-17 (+/- 8)	-12 (+/- 5)
1.3	-31 (+/- 6)	-14 (+/- 9)
1.9	-26 (+/- 3)	-15 (+/- 4)
2.1	-28 (+/- 19)	-16 (+/- 10)

Results:

Film Stress Evolution and Whisker Growth

- Increasing film stress correlates with an increased propensity to whisker
- Increased film stress correlates with decreased GB mobility
- What are the active stress relief mechanisms?

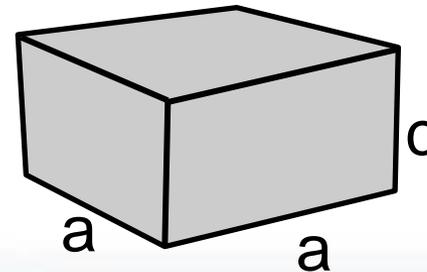
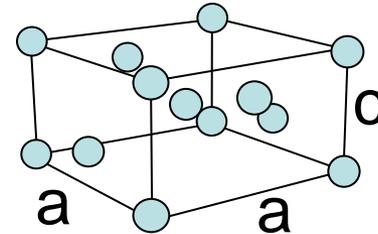


Film stress as a function of time for Sn and SnCu electroplated films.

Methods:

Microstructure - Crystallographic Texture

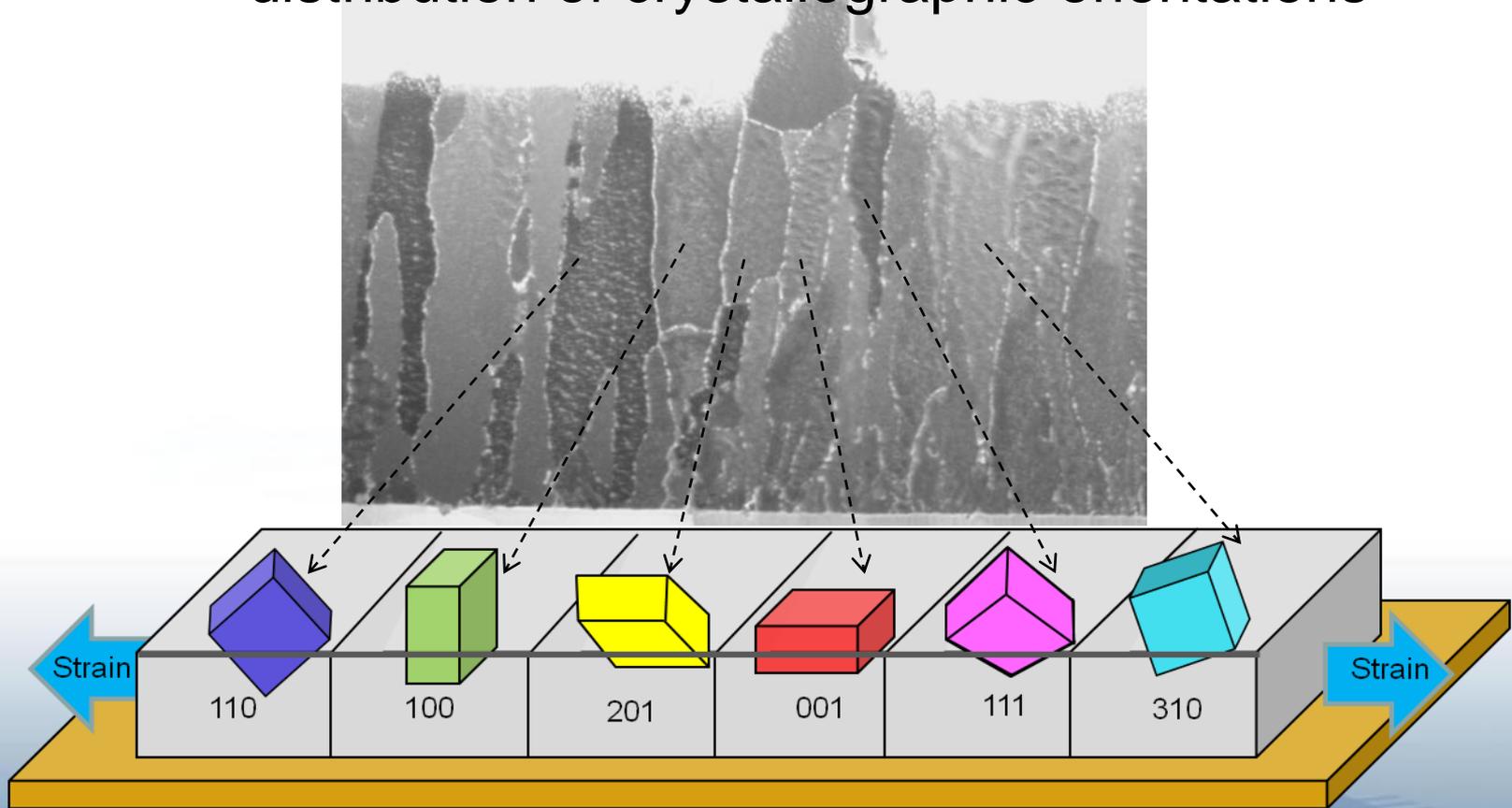
- Crystal Structure
 - Building block of atomic arrangements
- Sn Crystal Structure
 - Body-centered tetragonal (BCT)
 - $a = 0.583\text{nm}$
 - $c = 0.318\text{nm}$
- Crystallographic Texture
 - Distribution of crystallographic orientations



Methods:

Microstructure – Crystallographic Texture

Crystallographic texture describe the distribution of crystallographic orientations



Method:

Microstructure – Crystallographic Texture

- Observations by Arnold⁹
 - Observed that basic processing parameters influenced the propensity of an electroplated film to whisker
 - Current density
 - Electrolyte additives
 - Plating thickness
- These factors also influence crystallographic texture⁴
- Observations from Lee and Lee⁸
 - Focused on the mechanically anisotropic nature of Sn
 - For a stress of 8MPa, the expected strain could vary by 300%
 - Reaction to a strain affects the local stress state of a grain
 - Local stress gradient affects mass transport

Method:

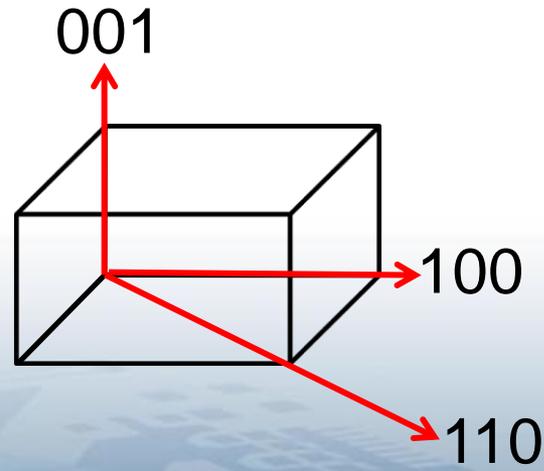
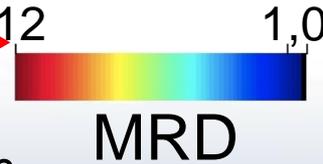
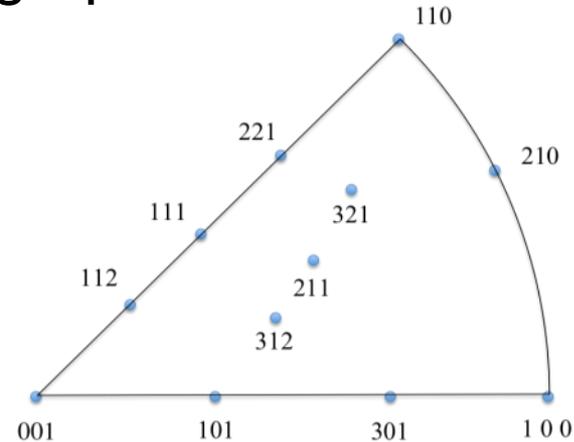
Microstructure – Crystallographic Texture

- Two commercial Sn electrolytes
 - Electrolyte 'A'
 - Reported to be whisker resistant
 - Electrolyte 'B'
 - Reported to be whisker prone
- Range in thickness
 - 1, 2, 4, 8, 12 μ m
- 40 or 80 mA/cm² deposition current density
- Measured texture the day after and 30 days after plating using XRD

Method:

Microstructure – Crystallographic Texture

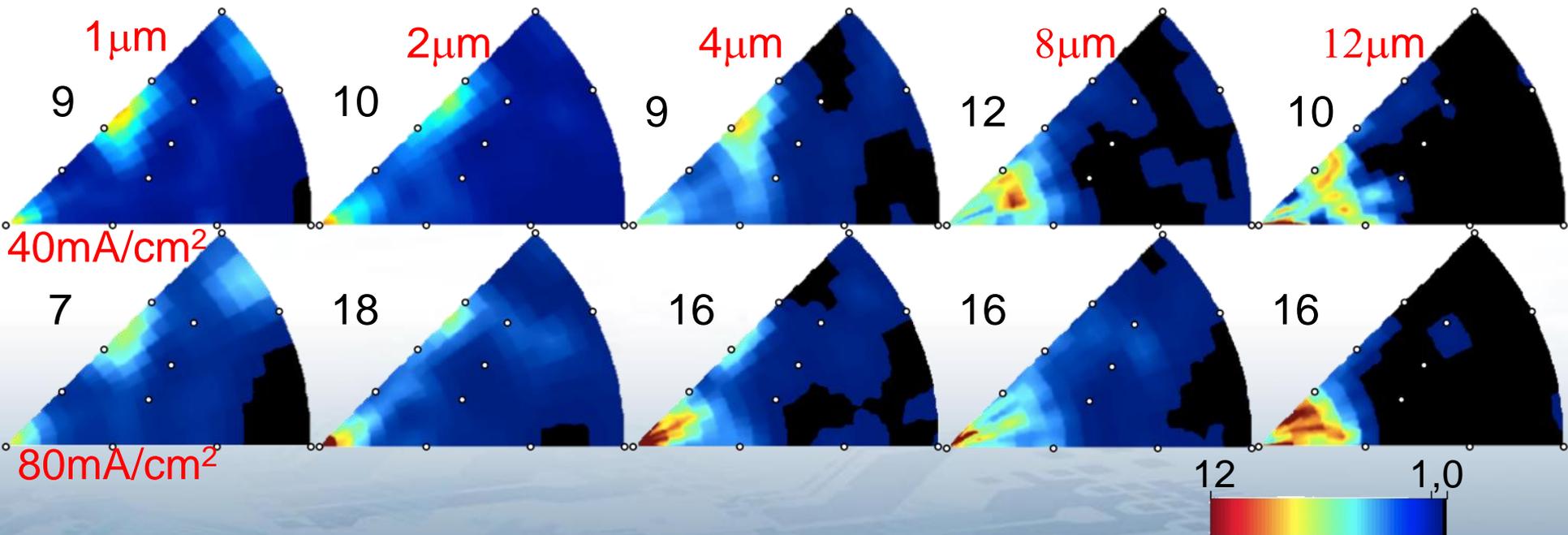
- Texture results presented as inverse pole figures
 - Topographical maps where the color represents the degree of texture in **multiple of random distribution (MRD)**
 - <1 is less probable
 - 1 = same as random distribution
 - >1 is more probable



Results:

Crystallographic Texture for 'A'

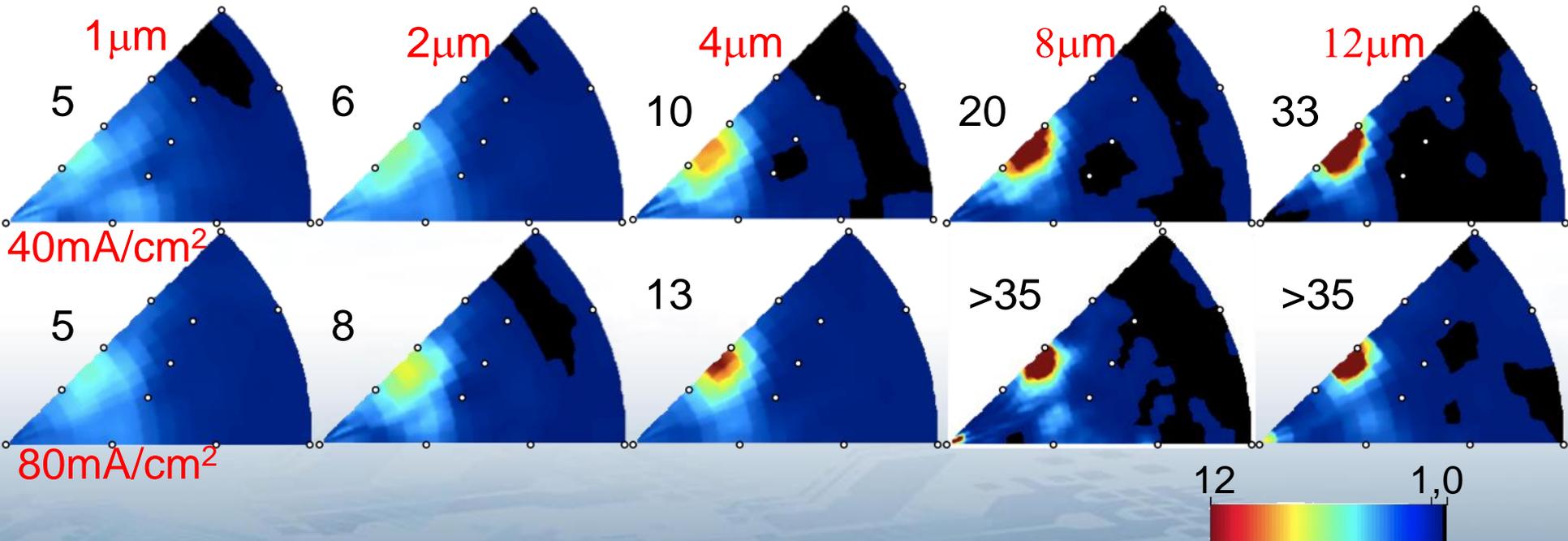
- Texture changes due to film thickness and deposition current density



Results:

Crystallographic Texture for 'B'

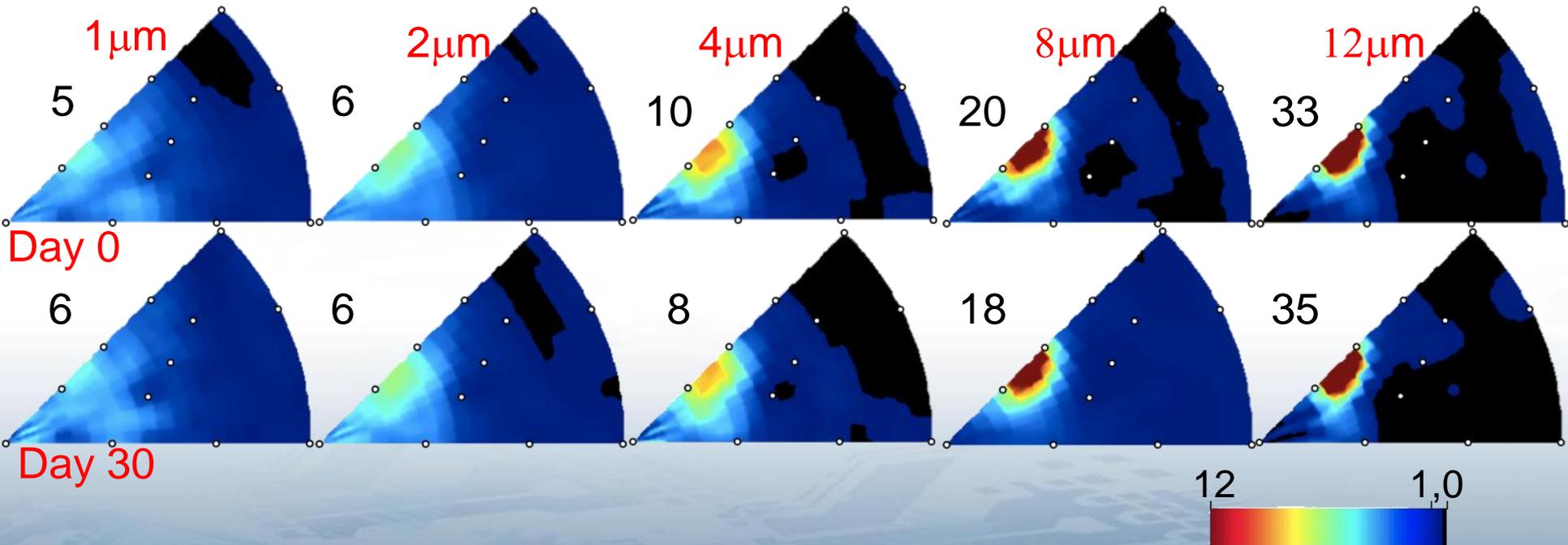
- Texture changes due to film thickness and deposition current density



Results:

Crystallographic Texture for 'B'; Day 30

- No hillock and whisker growth after 30 days
- No change in crystallographic texture

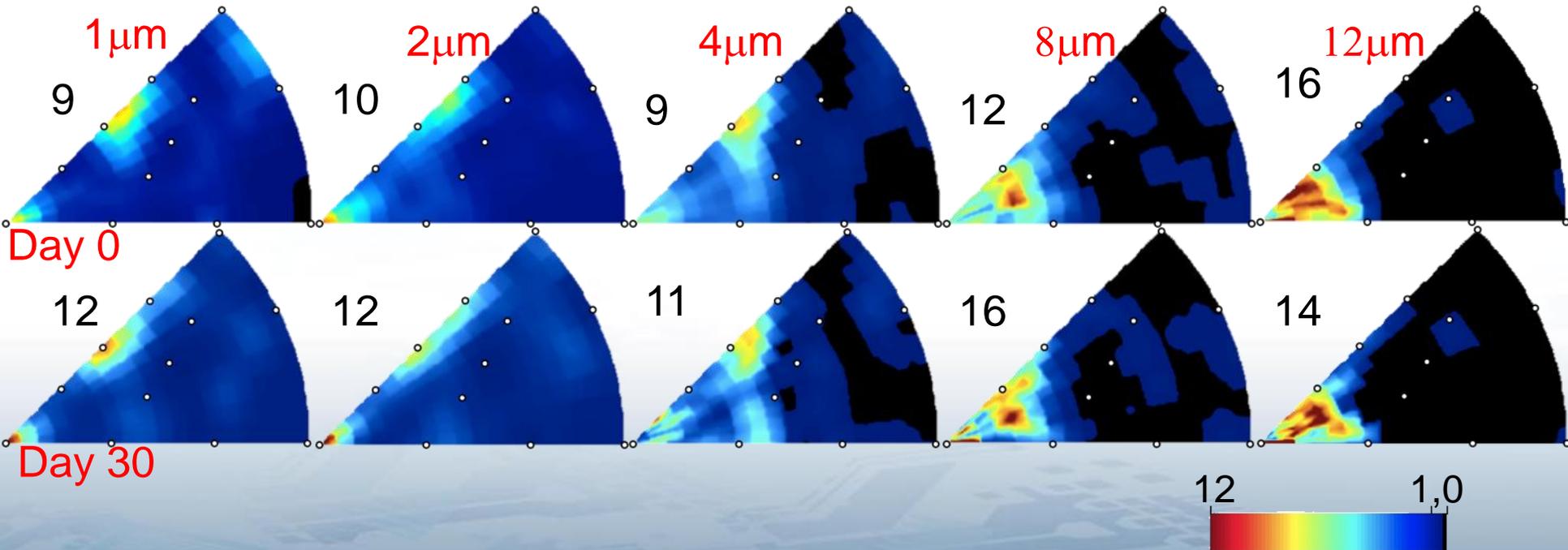


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Results:

Crystallographic Texture for 'A'; Day 30

- Hillock growth after 30 days on films 8 and 12 μ m thick
- Changes in crystallographic texture for all films



Results:

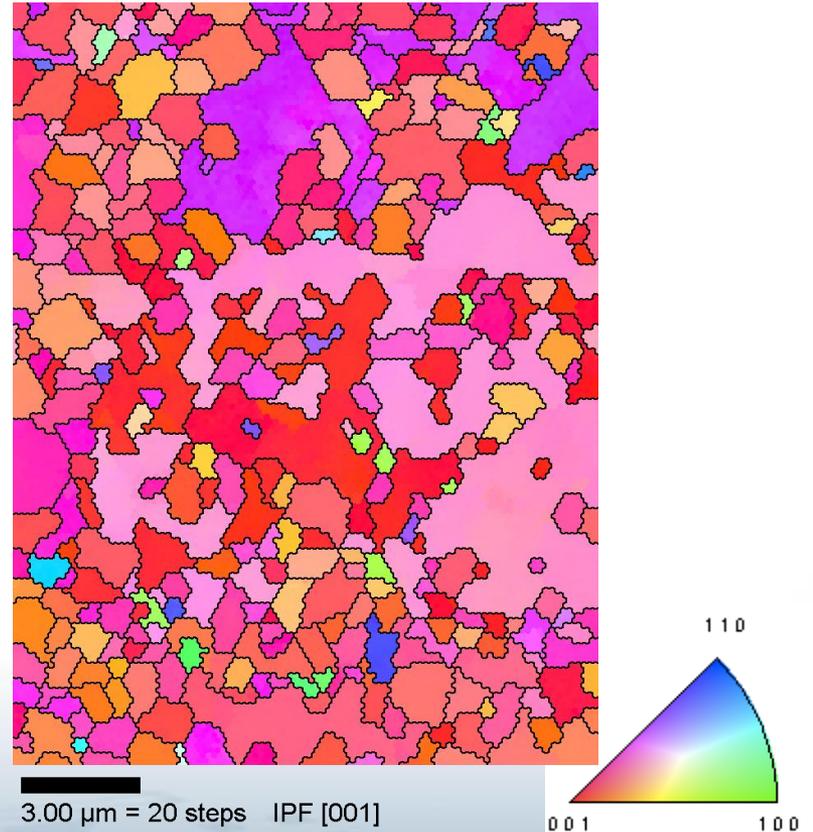
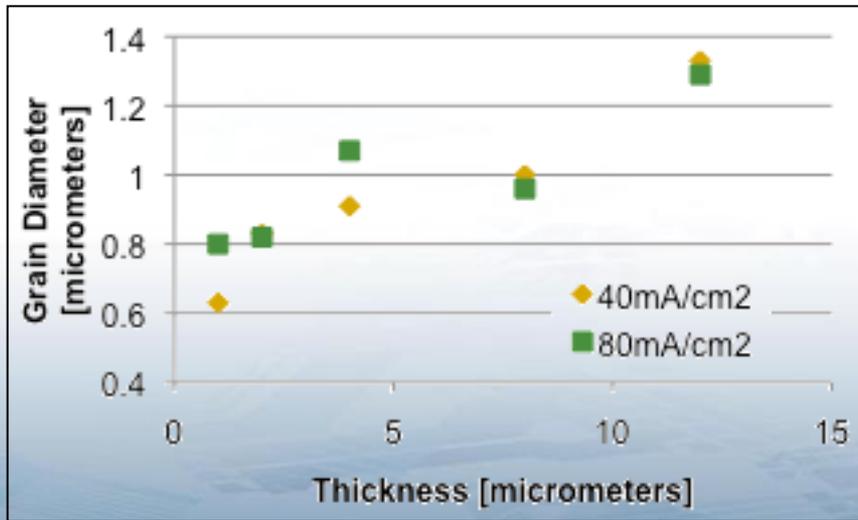
Crystallographic Texture Summary

- All processing parameters results in differences in crystallographic texture
 - Electrolyte type
 - Deposition current density
 - Film thickness
- Change in texture after 30 days for films deposited from electrolyte 'A' but not from electrolyte 'B'
- Evidence of recrystallization and/or grain growth with changing texture
 - Possible stress relief mechanism
- Increase in particular orientation correlated to hillock growth
 - Possible orientation dependence
 - Role of active slip systems (mechanism)

Results:

Grain Size Development – Electrolyte ‘A’

- Increasing grain diameter with thickness
 - Grain diameter between 0.6 and 1.4 μm
- Bimodal distribution of grain size
- No clear effect due to current density



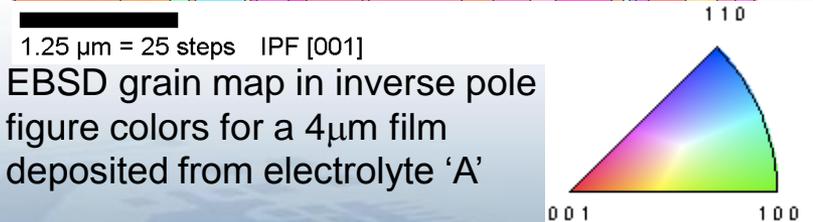
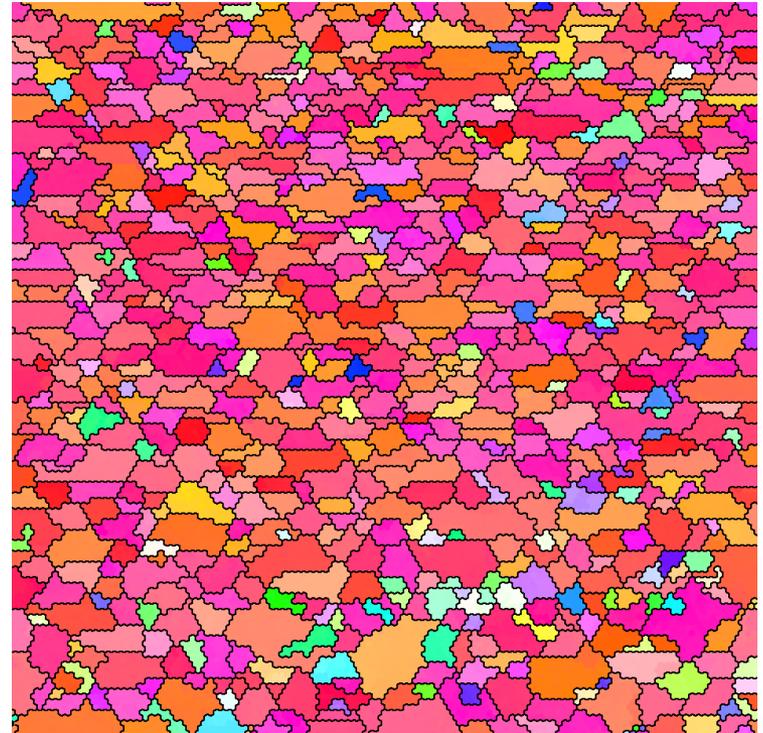
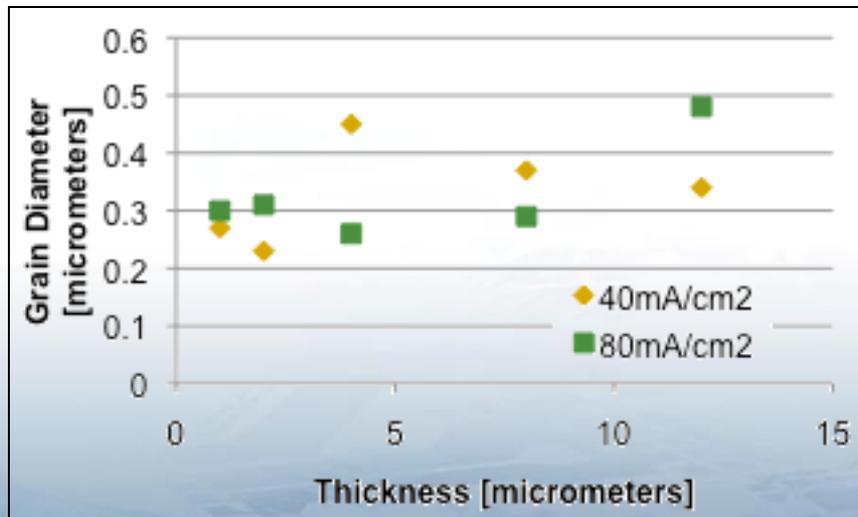
EBSD grain map in inverse pole figure colors for a 4 μm film deposited from electrolyte ‘A’

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Results:

Grain Size Development – Electrolyte ‘B’

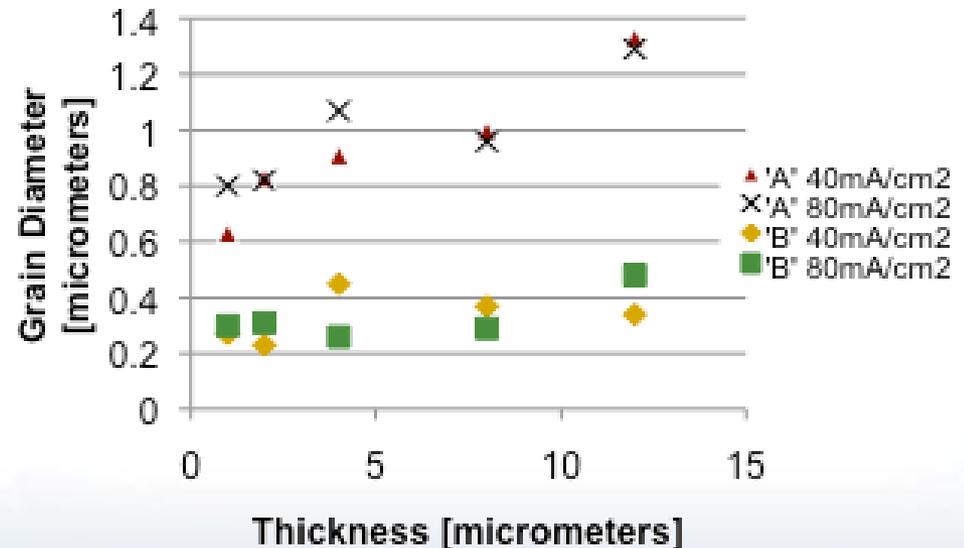
- Grain diameter between 0.2 to 0.5 μm
- No clear effect due to current density



Results:

Grain Size Summary

- Films deposited from electrolyte 'B' have a finer grain structure
- The grain size of films deposited from electrolyte 'A' increase with thickness but there is no clear effect with films from electrolyte 'B'
- Films from electrolyte 'A' have a bimodal distribution of grain diameters



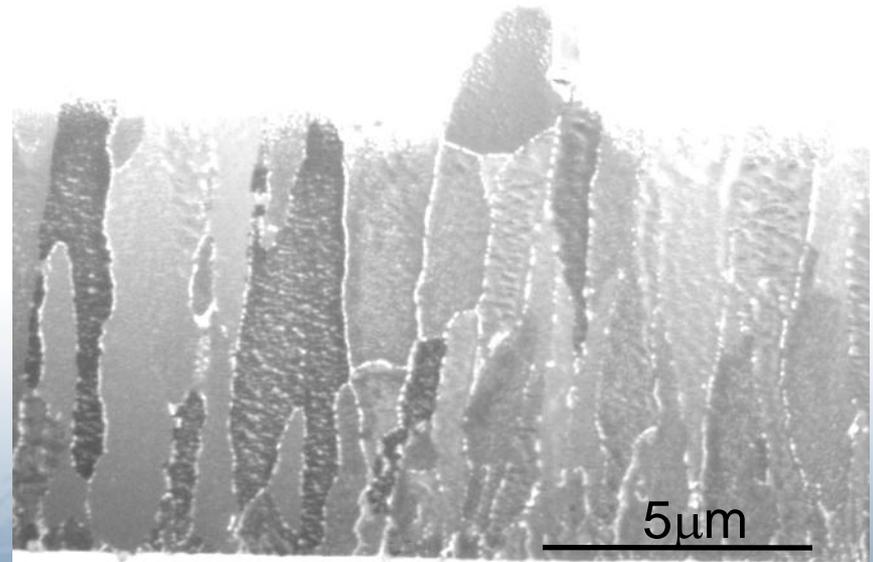
Results:

Microstructure – Grain Morphology – Electrolyte ‘A’

- Columnar grain growth observed at all thicknesses evaluated
- Existing grains terminate and new grains form throughout the thickness of film



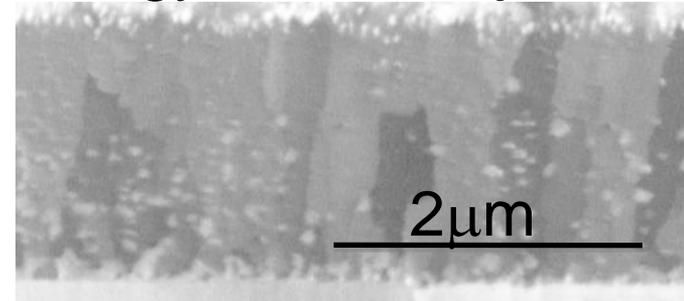
FIB cross section 4μm(above) and 12μm (below) thick film deposited from electrolyte ‘A’



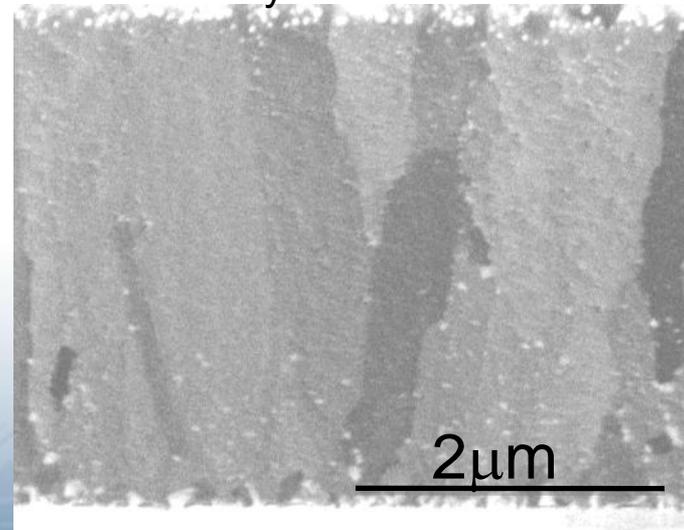
Results:

Microstructure – Grain Morphology – Electrolyte ‘B’

- Columnar grain growth observed at all thicknesses evaluated
- Existing grains terminate and new grains form throughout the thickness of film



FIB cross section 2μm (above) and 4μm (below) thick film deposited from electrolyte ‘B’



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Concluding Remarks

Correlating film properties to the propensity to whisker will lead to the development of a comprehensive whisker growth model and create successful mitigation strategies.

- Hillock and whisker growth have been correlated to measureable film properties
 - Film stress evolution
 - Grain boundary mobility
 - Crystallographic texture evolution
- Film properties are influenced by basic processing parameters
 - Electrolyte type
 - Electrolyte additives
 - Deposition current density
 - Film thickness

References

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9. S.M. Arnold, "The Growth of Metal Whiskers on Electrical Components", Proc.Elec. Components Conference, pp. 75-82, 1959.18

Questions

Improving Tin Whiskers through Quantitative Measurements of Plated Film Properties

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September 29th, 2010

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